

Computing the Magnetic Structure at Interfaces

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Magnetic Nanostructures, Interfaces, and New Materials
ALS, Berkeley, October 19, 2004

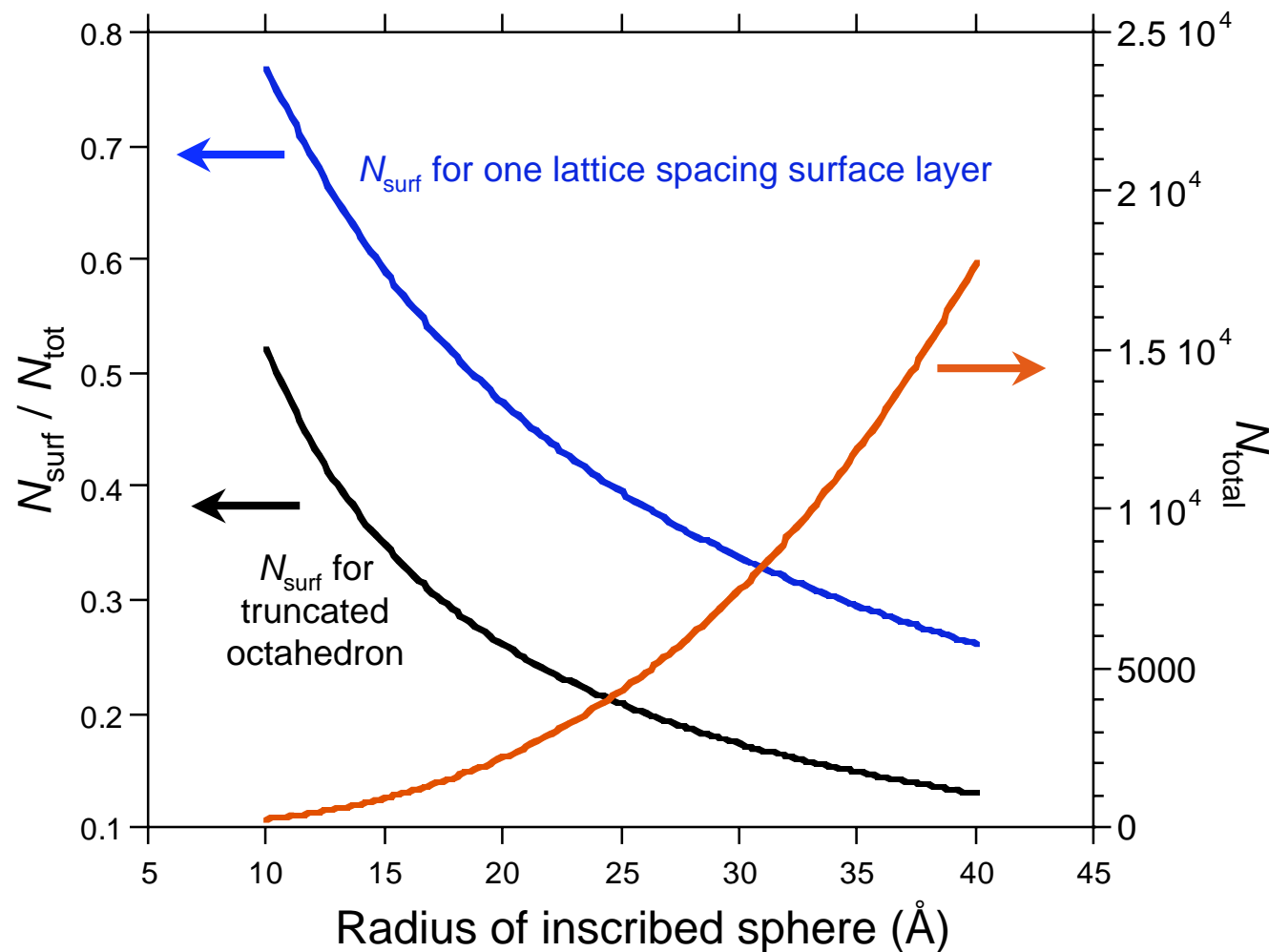
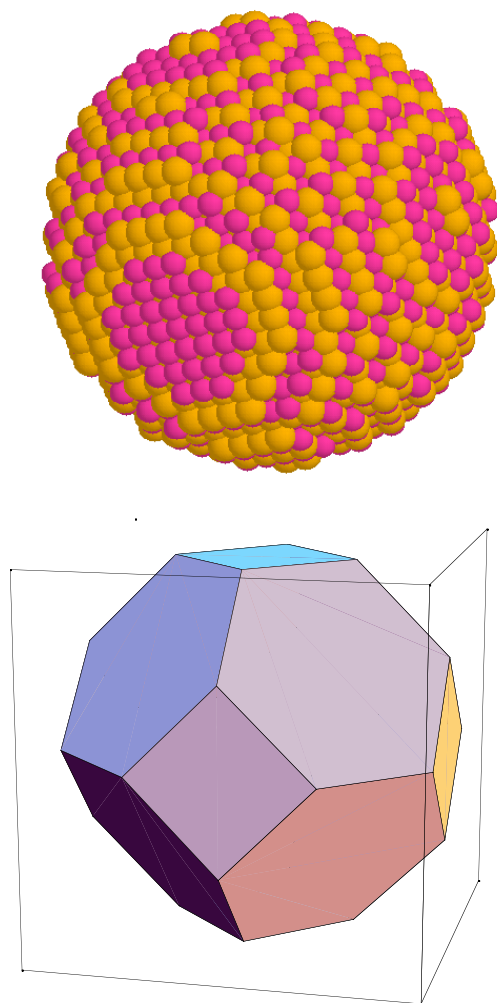
- **Outline:**

- **Magnetic nanostructures: opportunities & challenges**
- ***Ab initio* modeling of magnetic systems**
- **FeMn and FeMn|Co heterostructures**
- **Future plans**

- **Collaborators:**

- **Balazs Ujfalussy (ORNL & Budapest, Hungary)**
- **Malcolm Stocks (M&C Division, ORNL)**

Example: FePt (L10) Nanoparticle



⇒ **Nanoscale is dominated by interface**

Surface Spin Disorder in Nanoparticles

VOLUME 77, NUMBER 2

PHYSICAL REVIEW LETTERS

8 JULY 1996

Surface Spin Disorder in NiFe_2O_4 Nanoparticles

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Physics Department, University of California at San Diego, La Jolla, California 92093

E. J. McNiff, Jr. and S. Foner

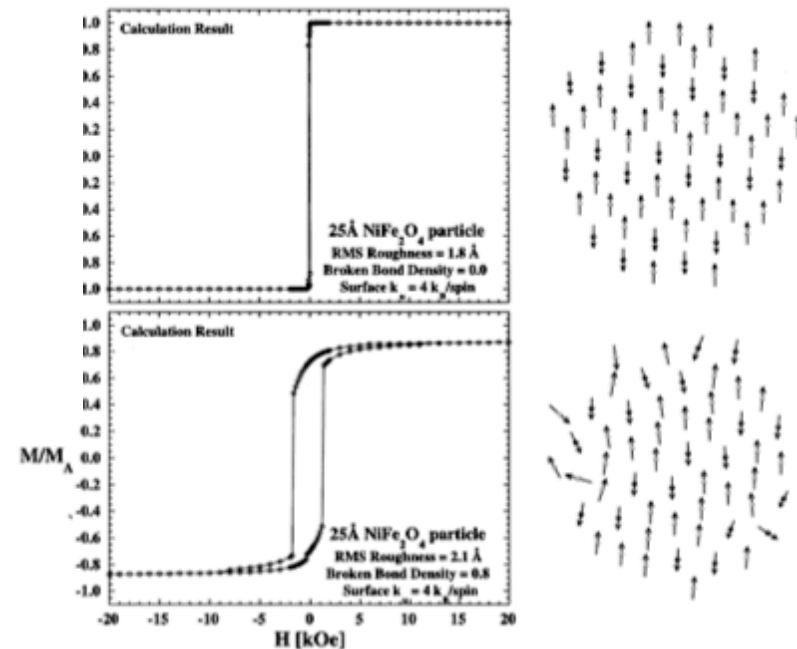
Francis Bitter National Magnet Laboratory, Cambridge, Massachusetts 02139

(Received 6 February 1996)

Nickel ferrite nanoparticles exhibit anomalous magnetic properties at low temperatures: low magnetization with a large differential susceptibility at high fields, hysteresis loops which are open up to 160 kOe, time-dependent magnetization in 70 kOe applied field, and shifted hysteresis loops after field cooling. We propose a model of the magnetization within these particles consisting of ferrimagnetically aligned core spins and a spin-glass-like surface layer. We find that qualitative features of this model are reproduced by a numerical calculation of the spin distribution. Implications of this model for possible macroscopic quantum tunneling in these materials are discussed. [S0031-9007(96)00628-X]

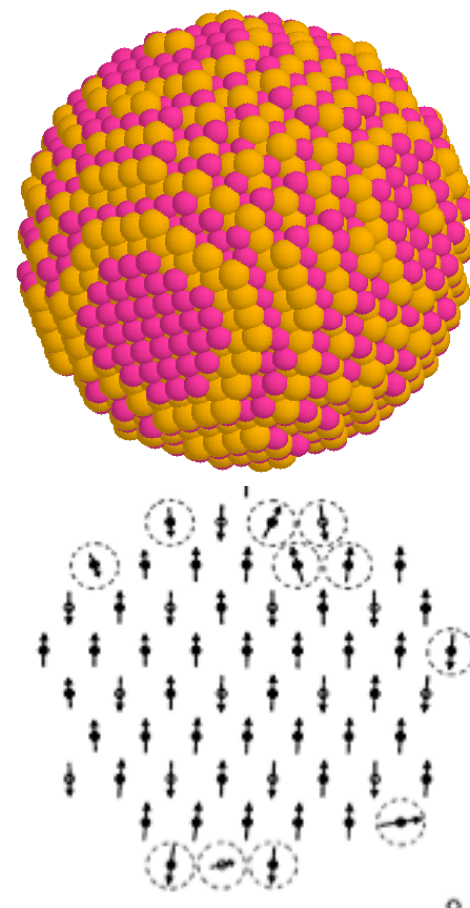
PACS numbers: 75.50.Tt, 75.30.Pd, 75.50.Gg, 75.50.Lk

- **Competition between surface anisotropy and exchange**
- **Maybe even random exchange**
- **Ferro- (Ferri-) magnetic interior couples to spin-glass (antiferromagnetic) surface region**
- **Enhancement of “effective” anisotropy**



Nanoparticles / Nanostructures

- Nano-particles/gains = “single domain”
- 6 nm particle ~40% of atoms in surface region
- Competition between chemical phases
 - Multiple magnetic phases
 - Surface segregation
- Demonstrated with Kodama’s work on NiFe_2O_4 :
magnetic structure probably complex!
- Interfaces between “materials” or magnetic phases
 - Ferro/antiferro; ferro/sin-glass; ...
 - Exchange bias was discovered in Co nanoparticles!
- Because of large surface to volume ratio
all interesting effects occur in one particle
- Interest / opportunities in magnetic systems is due to there complexity
 - Long range interaction; non-linearity of LLG; ...



**Complex spin structure in
nanostructures could be an opportunity**

Atomic scale spin-structure

- **Nanomagnetism, opportunities & challenges**
 - need to understand atomic scale spin structure
 - **Experimental probes**
 - Most average over 10-100 or more nanometers
 - Atomic scale spin-structure accessible by inference using models
 - **Models of magnetism at the nano scale**
 - Surface / interface region is like new material
 - Parameters (exchange, anisotropy, etc.) unknown
- ⇒ **Complement experiment with first principle electronic structure calculations**

Ab initio modeling of magnetism

Self Consistent Field Equations

$$\tilde{V}^{eff}[n, \mathbf{m}] = V^{ext} \tilde{\mathbf{I}} + \mathbf{B}^{ext} \cdot \tilde{\boldsymbol{\sigma}} + -\tilde{\mathbf{I}} \int d\mathbf{r}' n(\mathbf{r}) / |\mathbf{r} - \mathbf{r}'| \\ + \tilde{\mathbf{I}} \left(\frac{\delta \Omega_{xc}}{\delta n(\mathbf{r})} + \frac{\delta \Omega_{xc}}{\delta \mathbf{m}(\mathbf{r})} \right) \cdot \tilde{\boldsymbol{\sigma}}$$

LSDA : exchange/correlation

$$\{(\epsilon + \frac{\hbar}{2m} \nabla^2) \tilde{\mathbf{I}} - \tilde{V}^{eff}\} \tilde{\mathbf{G}}(\mathbf{r}, \mathbf{r}'; \epsilon) = \tilde{\mathbf{I}} \delta(\mathbf{r} - \mathbf{r}')$$

$$n(\epsilon) = \text{Im} \frac{1}{\pi} \int d\mathbf{r} \text{Tr} \tilde{\mathbf{G}}(\mathbf{r}, \mathbf{r}'; \epsilon)$$

$$n(\mathbf{r}) = \text{Im} \frac{1}{\pi} \int d\epsilon f(\epsilon - \mu) \text{Tr} \tilde{\mathbf{G}}(\mathbf{r}, \mathbf{r}'; \epsilon)$$

$$\mathbf{m}(\mathbf{r}) = \text{Im} \frac{1}{\pi} \int d\epsilon f(\epsilon - \mu) \tilde{\boldsymbol{\sigma}} \text{Tr} \tilde{\mathbf{G}}(\mathbf{r}, \mathbf{r}'; \epsilon)$$

$$\Omega = \int d\epsilon f(\epsilon - \mu) \epsilon n(\epsilon)$$

$$- \frac{e^2}{2} \iint d\mathbf{r} d\mathbf{r}' n(\mathbf{r}) n(\mathbf{r}') / |\mathbf{r} - \mathbf{r}'|$$

$$+ \Omega_{xc} - \int d\mathbf{r} \left\{ \frac{\delta \Omega_{xc}}{\delta n(\mathbf{r})} n(\mathbf{r}) + \frac{\delta \Omega_{xc}}{\delta \mathbf{m}(\mathbf{r})} \cdot \mathbf{m}(\mathbf{r}) \right\}$$

▪ Magnetic moments:

-Fe: ~2.3 μ_B exp. 2.2 μ_B

-Co: ~1.6 μ_B exp. 1.7 μ_B

-works for many other materials (Ni, Cr, etc.)

▪ Anisotropy: similar success

▪ Curie temperatures:

-Mean Field with Onsager Cavity Fields

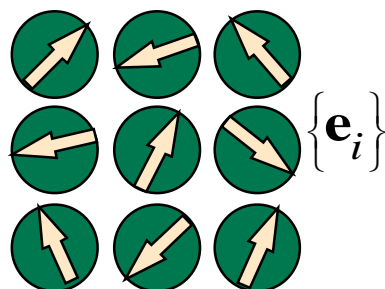
	Fe	Ni
T_c (K)	1015	450
Expt.	1040	631
m_{DLM} (μ_B)	1.89	0.0

Gyorffy, et al. *J. Phys. F*15, 1337 (1985); *ibid* 1761

Staunton & Gyorffy, *PRL* 69, 371 (1992)

Constrained Local Moment Method

- **Local moment: (even in itinerant transition metals)**



$$\mathbf{e}_i = \frac{\int_{\Omega_i} d\mathbf{r} \, \mathbf{m}_M^i(\mathbf{r})}{\left| \int_{\Omega_i} d\mathbf{r} \, \mathbf{m}_M^i(\mathbf{r}) \right|}$$

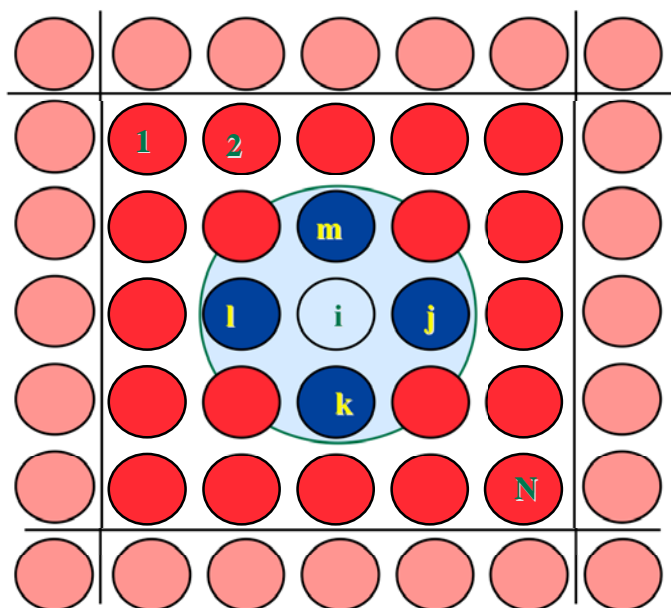
- **Time evolution of magnetic structure**

$$\frac{d}{dt} \mathbf{e}_i = -\frac{2\mu_B}{\hbar \bar{\mu}_i} [\mathbf{h}_i^{eff} \times \mathbf{e}_i]$$
$$\mathbf{h}_i^{eff} = \frac{\partial F}{\partial \mathbf{e}_i}$$

- **Classical dynamics but ab initio free energy functional**

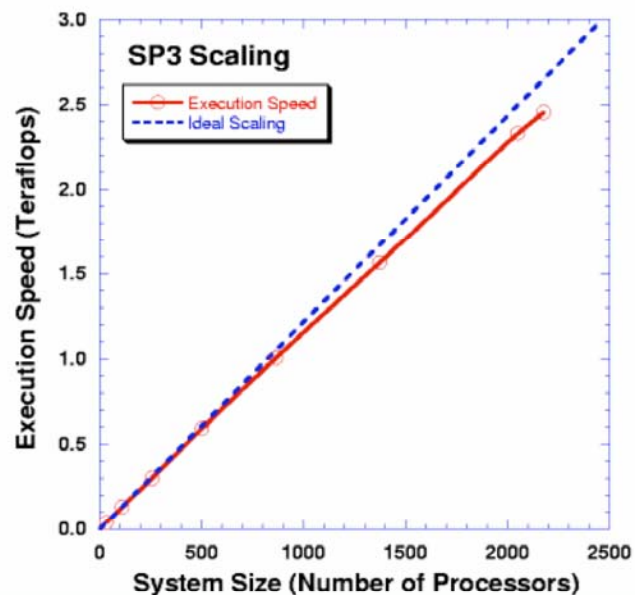
Locally Self-Consistent Method

- Moment / charge of atom i
 - Schrödinger eq. in cluster
 - Poisson eq. in solid



- Algorithm maps perfectly onto massively parallel computers

- FeMn runs on IBM SP3 at NERSC

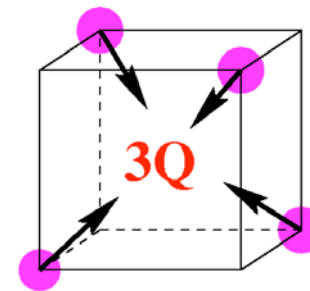
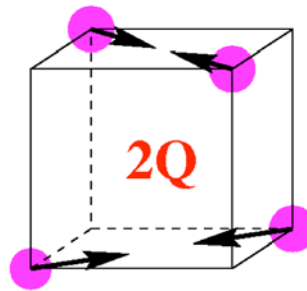
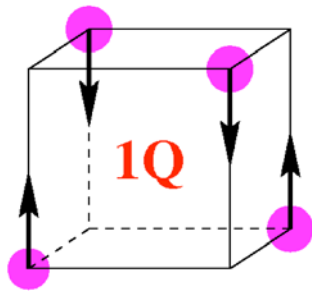


- Calculations with $\sim 10^4$ atoms are possible
 - 5nm^3 has $\sim 12,000$ atoms

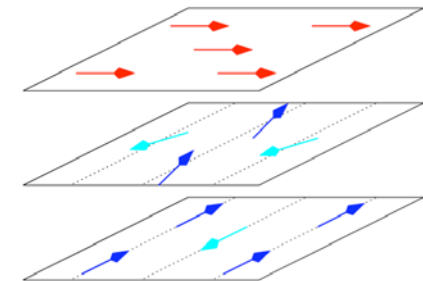
Magnetism in γ FeMn & FeMn/Py

Experimental Summary

- Inelastic Neutron Scattering: FeMn is antiferromagnetically ordered in 1Q, 2Q, and 3Q (Umebayashi and Ishikawa (1966))
- Mossbauer spectroscopy suggests 3Q or 2Q (Kennedy and Hicks (1987))



- But near neighbor Heisenberg model cannot distinguish between these three structures
- Perpendicular coupling between FeMn and Permalloy observed (Jungblut et. al. 1995)
 - understood in Heisenberg model (Hinchey & Mills 1986, Koon 1997)

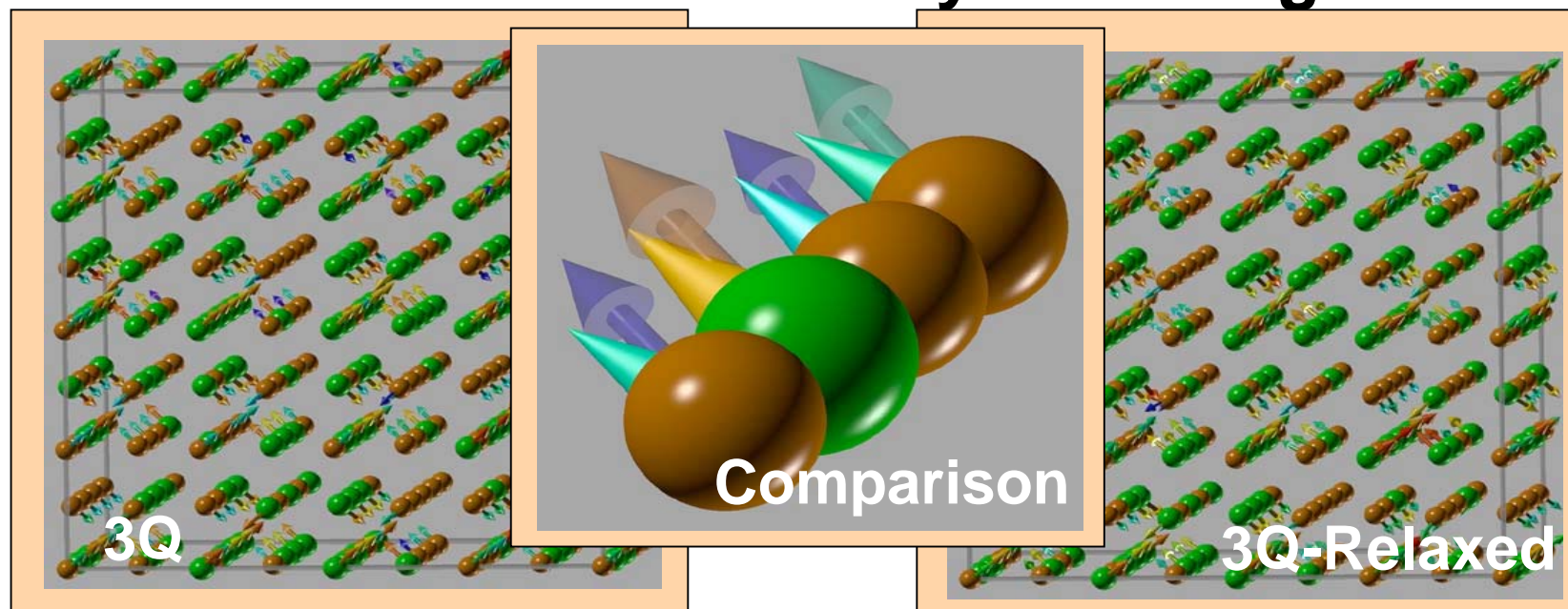


Ab initio calculations for bulk FeMn

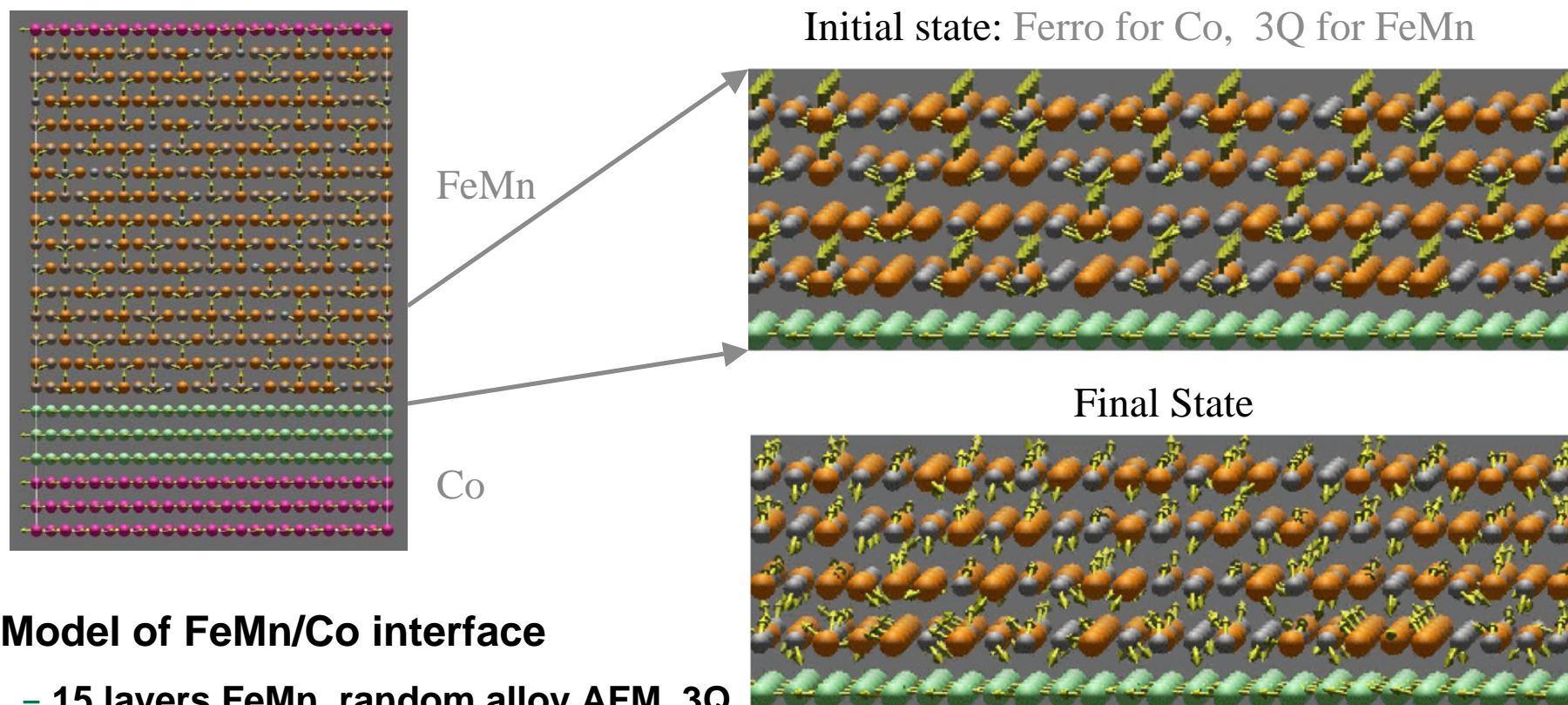
- In a nutshell: 1Q is unstable, 2Q & 3Q are stable**

Mean Field (CPA)				Real Space (super cell)				
KKR-CPA	1 Q	2Q	3Q	LSMS	1 Q	2Q	3Q	3Q_R
μ_{Mn}	1.88	2.00	2.05	μ_{Mn}	2.09	2.14	2.17	2.21
μ_{Fe}	1.72	1.85	1.91	μ_{Fe}	1.61	1.75	1.79	1.77
$E_{\text{B}}(\text{meV})$	15.86	5.27	0.0	$E_{\text{B}}(\text{meV})$	15.0	4.14	0.0	-2.5

- Moment direction varies only a few degrees**



FeMn | Co Multilayers

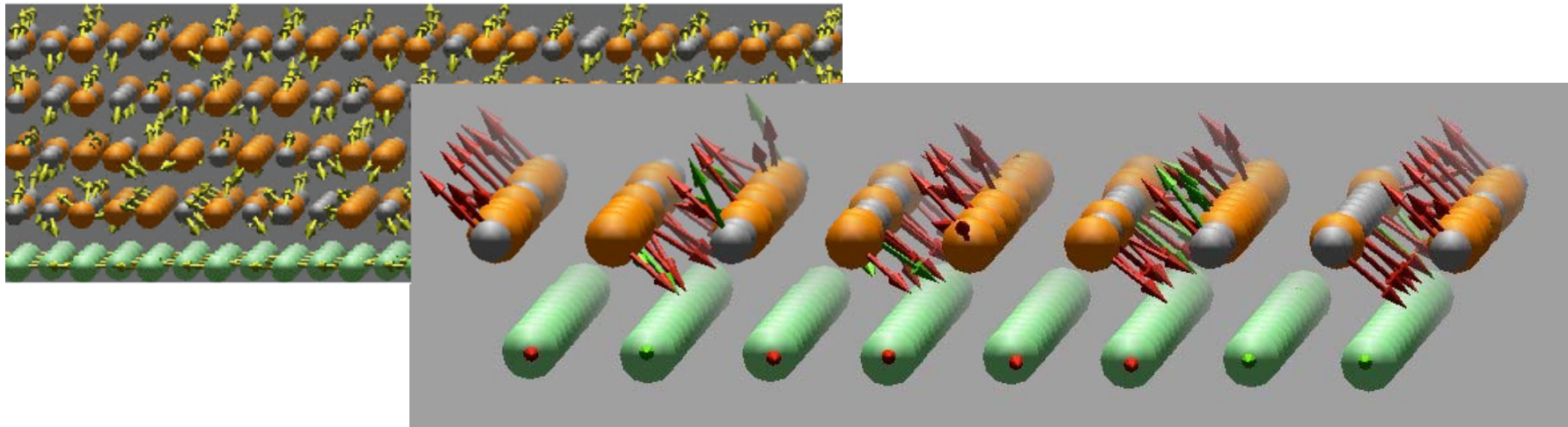


- **Model of FeMn/Co interface**

- 15 layers FeMn, random alloy AFM, 3Q
- 6 layers Co – 3-frozen FM
- Base 2D cell 12x8
- 2016-atom calculation

Co → Remains ferromagnetic
FeMn → 3Q structure destroyed

Rearranged spin structure due to proximity of FeMn to Co

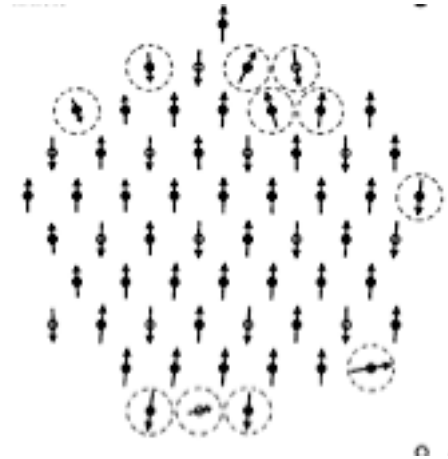


- **Proximity to Co:**
 - spin structure in FeMn changes from 3Q to 1Q
 - FeMn moments are approximately collinear
- **FeMn moments are perpendicular to Co**
 - Similar to spin-flop coupling in Heisenberg model

Looking into the Future

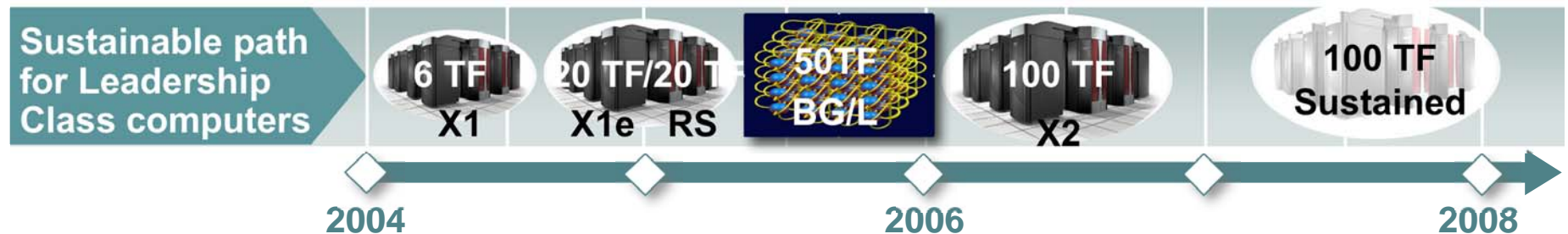
- In the next 3-5 years supercomputers will be 100-1000 times faster
 - Is it really reasonable to scale *ab initio* computation to 10^6 atoms?
- Nanomagnets require:
 - Ab initio calculations
 - Fluctuations at $T > 0$ - entropy effects
- Need to calculate free energy

$$F(\vec{e}, T) = E - T \ln W$$

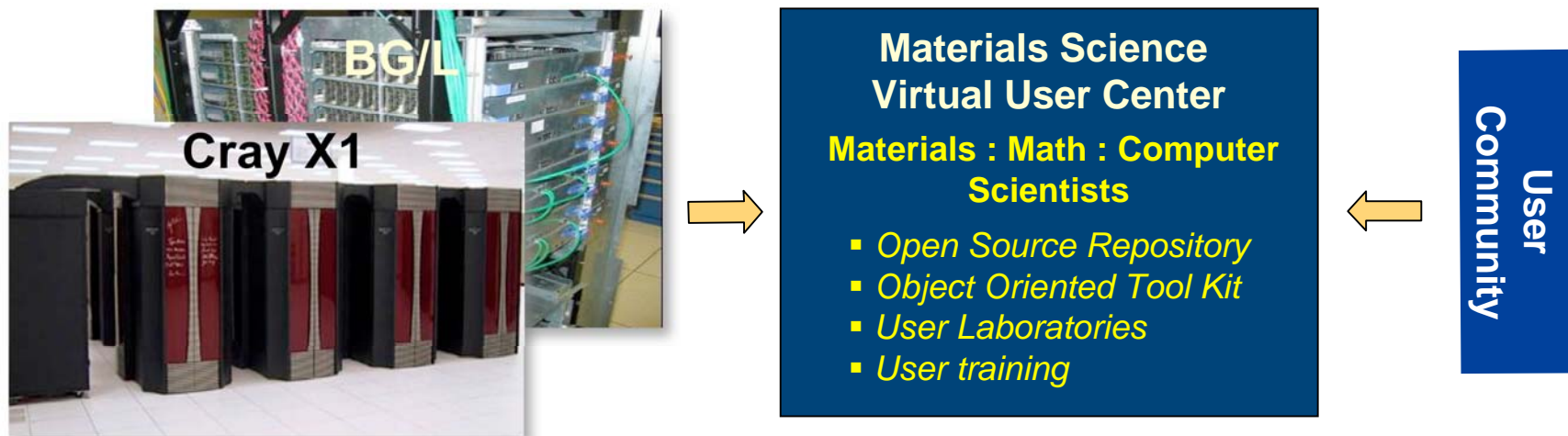


- Run multiple *ab initio* calculations in parallel to sample W
 - Probably doable for 100 times faster machine (ongoing project on IBM BG/L)
- Examples:
 - Temperature dependent effective anisotropy, magnetization, etc. in nanoparticles
 - Effective anisotropy for AFM/FM gains (Exchange Bias)

National Leadership Computing Facility



Instrumentation for Nanoscience



Computational End-Station for Nanomagnetism?

Summary and Conclusions

- **Reliable *ab initio* computation for nano-structures complement experimental probes**
 - Surface regions dominate nanostructures
 - Understand spin structure at the atomic scale
 - Materials specific studies
- **Applying *ab initio* computation to nanostructure is possible**
 - LSDA to DFT reliable for magnetic materials
 - Application to $\sim 10^4$ atom currently possible
- **Future possibilities
(3-5 years with 100-1000 x faster computers)**
 - Dynamics and finite temperature phenomena can be studied at atomic scale

Acknowledgment

This research used resources of the Center for Computational Sciences and was sponsored in part by the offices of Basic Energy Sciences and Advance Scientific Computing Research, U.S. Department of Energy. The work was performed at Oak Ridge National Laboratory, which is managed by UT-Battelle, LLC under Contract No. DE-AC05-00OR22725.